

# Use of L-tyrosine amino acid as biomodifier of Cloisite Na<sup>+</sup> for preparation of novel poly(vinyl alcohol)/organoclay bionanocomposites film

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**Abstract** Cloisite Na<sup>+</sup> was modified via cation exchange reaction using natural L-tyrosine amino acid. This novel chiral organo-modified Cloisite Na<sup>+</sup> was characterized using fourier transform infrared spectroscopy (FT-IR), X-ray diffraction (XRD), and thermogravimetric analysis/derivative thermal gravimetric (TGA/DTG). Then, polymer bionanocomposites were prepared by dispersing chiral organo-modified Cloisite Na<sup>+</sup> in poly(vinyl alcohol) (PVA) via ultrasonic irradiation. The novel bionanocomposites were characterized by FT-IR, UV-Visible, TGA, and XRD. Scanning electron microscopy and transmission electron microscopy were used to obtain the information about morphological structure of PVA/Cloisite Na<sup>+</sup>/Tyr bionanocomposites. The results showed that a mixture of intercalated and exfoliated Cloisite Na<sup>+</sup> dispersion in PVA matrix. The TGA data was compared with the pure PVA and the results showed that the introduction of a small amount of Cloisite Na<sup>+</sup>/Tyr led to improvement in thermal stability of the obtained new bionanocomposites. UV-Visible transmission spectra of pure PVA and new bionanocomposites film in the visible light region (400–800 nm) showed that they are rather transparent.

## Introduction

Recently, polymer nanocomposites are the subject of increased interest because of the unique properties that can be achieved with these materials. Owing to the nanoscale dispersion of the inorganic phase, very small filler content is enough to affect several properties of polymer matrix; thereby, their physical, thermal, and mechanical properties could be improved [1–5]. One of the nanofiller materials is nanolayer of montmorillonite (MMT). Nanocomposites consisting of these materials and organic polymers have suggested extreme investigation interests in recent times because their unique characteristics produce many potentially commercial applications. They can be transformed into novel materials possessing the advantages of both organic materials, such as light weight, flexibility, and good moldability, and inorganic materials, such as high strength, heat stability, and chemical resistance. Materials incorporating polymer/MMT hybrids can be able to achieve advanced degrees of stiffness, strength, and gas-barrier properties with the addition of small amount of inorganic substance [6–8].

MMT has medicinal properties and have been used for drug delivery and drug release systems [9–11]. It is composed of units made of two silica tetrahedral sheets centered with an alumina octahedral sheet, which is called 2:1 phyllosilicate that is able to forming stable suspensions in water. Its layers are stacked by weak dipolar or van der Waals forces, and it has both surface and edge charges. The charges on edges are simply available to modification [12–15]. To insure good compatibility between polymer and MMT in the preparation of nanocomposites, modification of the MMT is necessary. This modification causes an increase in the *d*-spacing, which facilitates the entry of the polymer into the clay gallery. This is usually done by

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cationic exchange using alkylammonium salts [16–20]. Amino acids possess the potential to be used as modifiers, which can increase the interlayer spacing of clay and improve its biodegradability [21, 22].

Poly(vinyl alcohol) (PVA) is highly biocompatible, non-toxic, and a water-soluble polymer [23–25]. This polymer has many applications in textile industry, membrane manufacturing, medical, cosmetic, food, packaging industries, and paper coating, because of its high tensile, impact and abrasion resistance, high biocompatibility and high barrier properties [26–29]. Owing to wide range applications of PVA, many efforts have been done to improve the properties of this polymer. The performances of PVA nanocomposite films depend on the properties of nanofillers, preparation methods, and the final nanostructures. Nanoclays such as MMT, is shown to have a significant effect on the properties of PVA nanocomposites which are attributed to the strong interactions between the nanoparticle and polymer matrix [30]. PVA/clay nanocomposites have been studied in many laboratories [31–33]. Sammon and co-workers [33] prepared PVA/MMT nanocomposites using solution intercalation method and performed in situ diffusion experiments of water and acetone into nanocomposites. Strawhecker and Manias [12] prepared the PVA/clay nanocomposites by casting the water suspension of PVA and MMT. Room temperature distilled water was used to form a suspension of  $\text{Na}^+$ -MMT. The TEM photograph of 20 wt% clay containing nanocomposite reveals the co-existence of silicate layers in the intercalated and exfoliated states.

In this investigation, at first, protonated form of L-tyrosine amino acid (Tyr) was used as biomodifier for modification of Cloisite  $\text{Na}^+$ , which can increase the interlayer spacing of Cloisite  $\text{Na}^+$  and also improve its biodegradability. Intercalation of amino acid between nanolayer of Cloisite  $\text{Na}^+$  was studied by different techniques. Then, PVA/Cloisite  $\text{Na}^+$ /Tyr bionanocomposites (BNCs) with different compositions were fabricated using ultrasonic irradiation. Furthermore, the films of these BNCs were prepared via solution casting method and were fully characterized by different methods.

## Experimental

### Materials

PVA  $-\text{[(CH}_2\text{CHOH)]}_n-$  used in this study was purchased from Merck Chemical Co., Mw = 145,000 g/mol with a degree of hydrolysis of 98%. L-Tyrosine ( $\text{C}_9\text{H}_{11}\text{O}_3\text{N}$ , Mw = 181.19 g/mol) and hydrochloric acid (HCl, 37%) were purchased from Merck Chemical Co. Cloisite  $\text{Na}^+$  was obtained from southern clay products, Gonzales, Texas

(USA) with a cation exchange capacity (CEC) of 92.6 mequiv/100 g of Cloisite  $\text{Na}^+$ .

### Preparation of the chiral organo-modified Cloisite $\text{Na}^+$

Ion exchange method was used to prepare organo-modified Cloisite  $\text{Na}^+$ . At first Tyr was placed in a beaker, followed by the addition of concentrated hydrochloric acid and 50 mL of deionized water. This solution was heated at 80 °C for 3 h. In another beaker, Cloisite  $\text{Na}^+$  was dispersed in 150 mL of deionized water for 3 h. The both solutions were mixed, and the resulting mixture vigorously was stirred at 60 °C for 6 h. The solution containing the organo-modified Cloisite  $\text{Na}^+$  was filtered warmly and washed with hot deionized water to remove chloride anions. The organically modified Cloisite  $\text{Na}^+$  was dried at 60 °C for 6 h. The dried precipitates was ground and screened with a 325-mesh sieve.

### Preparation of PVA/Cloisite $\text{Na}^+$ /Tyr BNCs film

PVA/Cloisite  $\text{Na}^+$ /Tyr BNCs film with different Cloisite  $\text{Na}^+$ /Tyr content (5, 10, and 15%) were fabricated by using the solution casting method. In a typical procedure, for each composition, 1 g of PVA was dissolved in 20 mL of deionized water at 90 °C under continuous stirring for at least 30 min. Organo-modified Cloisite  $\text{Na}^+$  solutions ( $>1$  wt%) were obtained by suspending a measured amount of Cloisite  $\text{Na}^+$ /Tyr (5, 10, and 15% vs. PVA, respectively) in deionized water at 40 °C under stirring for 3 h and sonicated for 30 min (using a MISONIX ultrasonic liquid processor, XL-2000 SERIES. Ultrasound was a wave of frequency  $2.25 \times 10^4$  Hz and power of 100 W). Both solutions are mixed, stirred for 6 h at 90 °C, and again sonicated for 30 min. The mixture was poured into a Petri dish. These films were dried at 40 °C for 24 h.

### Characterizations

#### Infrared spectroscopy (FT-IR)

Spectra of Cloisite  $\text{Na}^+$ , Cloisite  $\text{Na}^+$ /Tyr, and Tyr samples were recorded from KBr pellets. Spectra of the blend films of PVA/Cloisite  $\text{Na}^+$ /Tyr were recorded with a Jasco-680 FT-IR spectrophotometer (Japan) in transmittance mode from 400 to 4000  $\text{cm}^{-1}$  with resolution of 4  $\text{cm}^{-1}$ .

#### X-ray diffraction (XRD)

The XRD measurements were performed for the Cloisite  $\text{Na}^+$ , organo-modified Cloisite  $\text{Na}^+$ , and the BNC films using a Bruker, D8ADVANCE (Germany). Ni-filtered  $\text{CuK}\alpha$  radiation utilized at 45 kV and 100 mA. Wavelength

of emitted radiation is  $1.54056 \text{ \AA}$ . The samples were scanned in the  $2\theta$  mode using a step scanning method with a step-width of  $0.02^\circ/\text{min}$  in the range from  $2\theta = 1.4^\circ$  to  $10^\circ$ .

#### Thermal gravimetric analysis (TGA)

The TGA of Cloisite  $\text{Na}^+$ , Cloisite  $\text{Na}^+/\text{Tyr}$ , pure PVA and BNC films were carried out on STA503 Win TA instrument at a heating rate of  $10^\circ\text{C}/\text{min}$  under flowing nitrogen from room temperature to  $800^\circ\text{C}$ . All samples were dried under vacuum for 4 h at  $60^\circ\text{C}$  prior to be tested.

#### UV–Visible spectroscopy

The UV–Visible transmission spectra of BNC films were measured using JASCO V-750 UV–Visible spectrometer in the spectra range from 200 to 800 nm.

#### Scanning electron microscopy (SEM)

The morphology of PVA/Cloisite  $\text{Na}^+/\text{Tyr}$  BNC film was observed by means of scanning electron microscope (XL30, Philips) at an accelerating voltage of 5 kV.

#### Transmission electron microscopy (TEM)

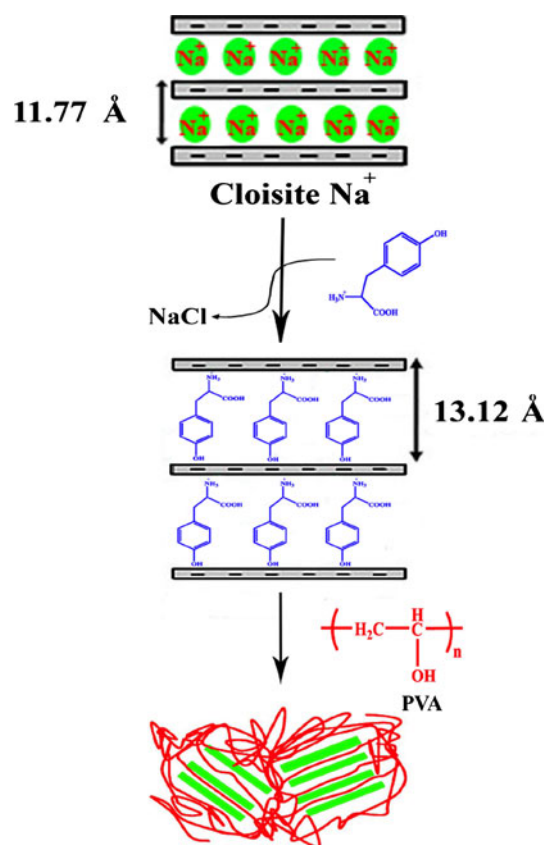
TEM micrograph of PVA/Cloisite  $\text{Na}^+/\text{Tyr}$  BNC was obtained with a Philips CM 120 model at an accelerating voltage of 100 kV.

## Results and discussion

#### Preparation and characterization of organoclay

Chiral organo-modified Cloisite  $\text{Na}^+$  was prepared by replacing  $\text{Na}^+$  cation in this nanoclay with protonated form of Tyr. This modification increased organophilicity of Cloisite  $\text{Na}^+$ , therefore dispersion of this nanoclay in polymer matrix was improved (scheme 1). The chemical structure, diffraction pattern, and thermal behavior of this organoclay were verified by FT–IR, XRD, and TGA/DTG.

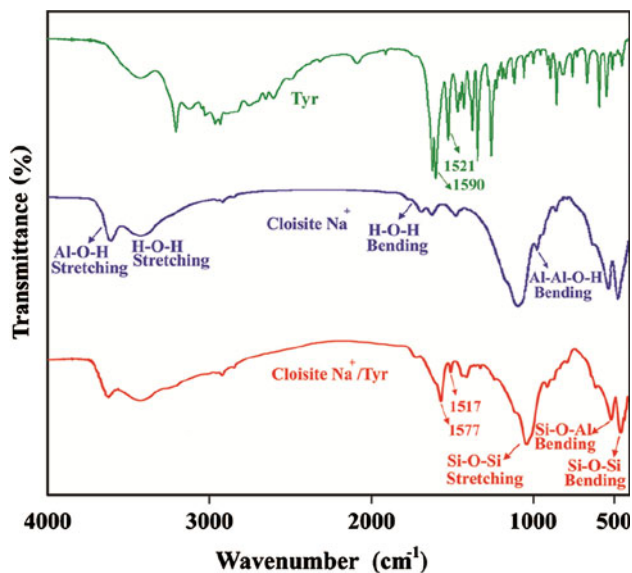
The FT–IR spectra of Tyr, Cloisite  $\text{Na}^+$ , and Cloisite  $\text{Na}^+/\text{Tyr}$  are shown in Fig. 1. The spectrum of Cloisite  $\text{Na}^+$  shows the characteristic bands  $3626 \text{ cm}^{-1}$  due to O–H stretching (sharp peak) for Al–OH and Si–OH, a broad peak centered on  $3441 \text{ cm}^{-1}$  due to interlayer and H-bonded –OH,  $1637 \text{ cm}^{-1}$  is attributed to –OH bending mode in water,  $1044 \text{ cm}^{-1}$  due to Si–O stretching,  $918$  and  $798 \text{ cm}^{-1}$  due to AlAlOH and AlMgOH bending vibrations and  $525$  and  $468 \text{ cm}^{-1}$  due to the Si–O–Al and Si–O–Si bending vibrations. In the spectrum of Tyr, broad peak which was



**Scheme 1** Schematic representation of the modification of Cloisite  $\text{Na}^+$  with L-Tyr and its subsequent use to produce BNCs

observed between  $3203$  and  $3125 \text{ cm}^{-1}$  corresponds to N–H and acidic O–H stretching, at  $2960$  and  $2752 \text{ cm}^{-1}$  which correspond to C–H stretching of alkyl chain. The peaks which were observed at  $1609$ ,  $1590$ ,  $1521$ , and  $1416 \text{ cm}^{-1}$  could be attributed to N–H asymmetric bending vibration, C=O asymmetric stretching, N–H symmetric bending vibration and C=O symmetric stretching, respectively. Cloisite  $\text{Na}^+/\text{Tyr}$  spectrum shows small broad peak between  $3300$  and  $2700 \text{ cm}^{-1}$  compare with the spectrum of pure Cloisite  $\text{Na}^+$  which could be attributed to acidic OH of Tyr. On the other hand, N–H bending and C=O stretching related to Tyr which were observed at  $1577$  and  $1517 \text{ cm}^{-1}$  showed small shift due to hydrogen bonding of protonated form of Tyr with Cloisite  $\text{Na}^+$ . All of this data revealed the presence of Tyr moieties in the Cloisite  $\text{Na}^+/\text{Tyr}$  system.

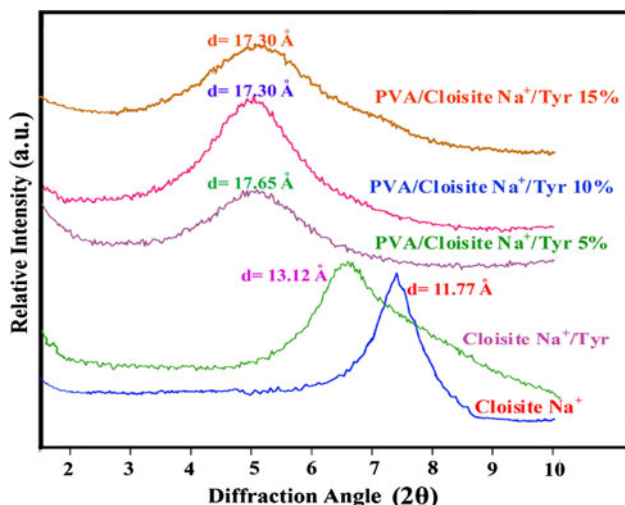
The XRD patterns of Cloisite  $\text{Na}^+$ , Cloisite  $\text{Na}^+/\text{Tyr}$  are shown in Fig. 2. The peak corresponding to pristine Cloisite  $\text{Na}^+$  is observed at  $2\theta = 7.5^\circ$ . The  $d$ -spacing at this value of  $2\theta$  is  $11.77 \text{ \AA}$ . In the case of Cloisite  $\text{Na}^+$  modified with Tyr, the peak appears at  $2\theta = 6.6^\circ$ , which related to a  $d$ -spacing of  $13.12 \text{ \AA}$ . The interlayer spacing of Cloisite  $\text{Na}^+$  was obviously increased after the treatment with Tyr. The result suggested that this novel biomodifier successfully



**Fig. 1** FT-IR spectra of Tyr, Cloisite Na<sup>+</sup>, and Cloisite Na<sup>+</sup>/Tyr

intercalated between layers of Cloisite Na<sup>+</sup>. The results of the XRD patterns are summarized in Table 1.

The TGA/DTG curves for Cloisite Na<sup>+</sup> and organo-modified Cloisite Na<sup>+</sup> are shown in Fig. 3a, c. Decomposition of Cloisite Na<sup>+</sup> occurs in two steps: desorption of water from the interlayer space happened around 100–200 °C which is about 1% weight loss and dehydroxylation of the layer crystal lattice structure took place around 700 °C Cloisite Na<sup>+</sup> [34]. The thermal degradation of Cloisite Na<sup>+</sup>/Tyr show three weight loss stages: the first weight loss occurs around 100 °C due to water trapped in the Cloisite Na<sup>+</sup>/Tyr chiral organoclay. The second step corresponds to decomposition of the organic parts present in the Cloisite Na<sup>+</sup>/Tyr system with a maximum



**Fig. 2** XRD patterns of Cloisite Na<sup>+</sup>, Cloisite Na<sup>+</sup>/Tyr, and different PVA/Cloisite Na<sup>+</sup>/Tyr BNCs

**Table 1** *d*-Value of Cloisite Na<sup>+</sup>, modified Cloisite Na<sup>+</sup> and PVA/Cloisite Na<sup>+</sup>/Tyr BNCs with different Cloisite Na<sup>+</sup>/Tyr contents

Sample	2θ (degree)	<i>d</i> (Å)
Cloisite Na <sup>+</sup>	7.5	11.77
Cloisite Na <sup>+</sup> /Tyr	6.6	13.12
PVA/Cloisite Na <sup>+</sup> /Tyr 5%	5.0	17.65
PVA/Cloisite Na <sup>+</sup> /Tyr 10%	5.1	17.30
PVA/Cloisite Na <sup>+</sup> /Tyr 15%	5.1	17.30

decomposition around 290–300 °C and third decomposition occurred around 600 °C.

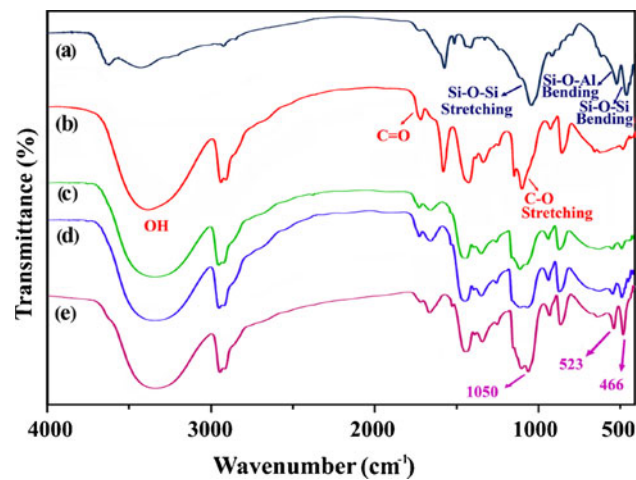
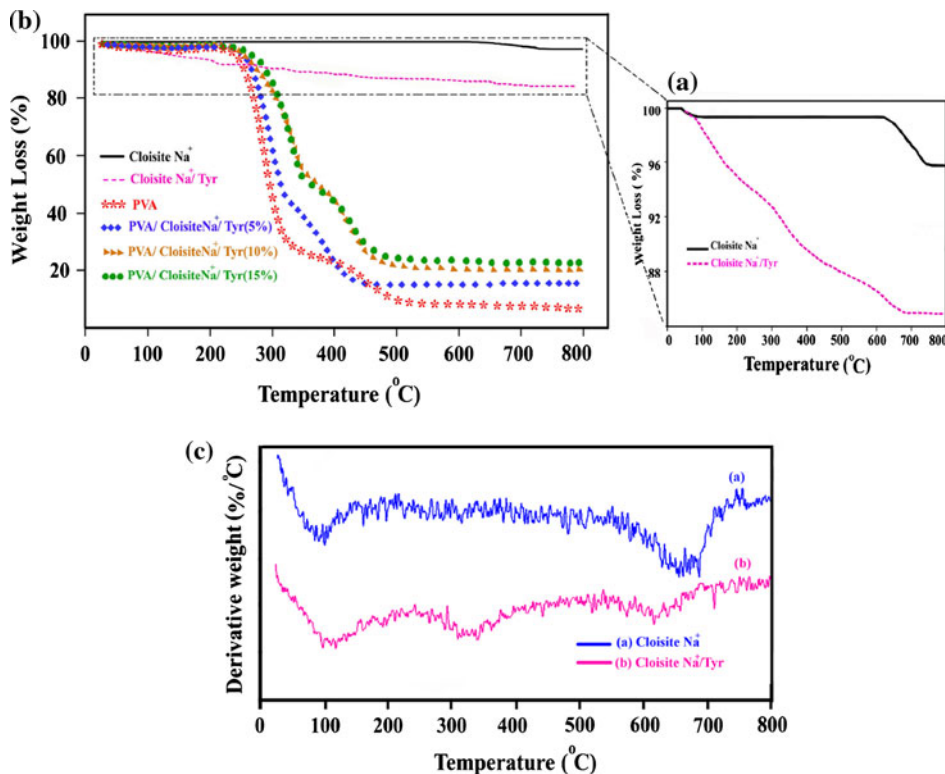
#### Preparation and characterization of PVA/Cloisite Na<sup>+</sup>/Tyr bionanocomposite films

PVA/Cloisite Na<sup>+</sup>/Tyr BNCs with different compositions were fabricated using ultrasonic irradiation. Furthermore, the films of these BNCs were prepared via solution casting method. The schematic preparation of BNC is shown in scheme 1. The BNCs were characterized by FT-IR, XRD, SEM, TEM, TGA, and UV-Visible, respectively.

FT-IR spectra pure PVA and PVA/Cloisite Na<sup>+</sup>/Tyr BNCs (5, 10, and 15% of Cloisite Na<sup>+</sup>/Tyr) are shown in Fig. 4. The frequencies and assignments for the pure PVA (Fig. 4b) are indicated as follows: very intensive, broad hydroxyl band occurs at 3000–3400 cm<sup>-1</sup>, bands at 2941 cm<sup>-1</sup> are due to stretching vibrations of -CH and CH<sub>2</sub> groups, resulting of residual acetate groups, is depicted at 1713 cm<sup>-1</sup> in PVA spectrum. A band attributed to C-O stretching is presented at 1093 cm<sup>-1</sup>. All these bands are also observed in BNCs film with Cloisite Na<sup>+</sup>/Tyr (Fig. 4c–e) which indicates the dispersion of chiral organoclay into PVA matrix. The very small shifts of absorption maximum are due to changes in the nearest surrounding of functional groups. As the loading of Cloisite Na<sup>+</sup>/Tyr is increased, the intensities of organoclay bands become stronger in the FT-IR spectra of the obtained BNC films.

Figure 2 shows the diffraction patterns of PVA/Cloisite Na<sup>+</sup>/Tyr BNCs with 5, 10, and 15% contents. PVA/Cloisite Na<sup>+</sup>/Tyr BNC 5% showed a peak at 2θ = 5.0° (*d* = 17.65 Å), for 10 and 15%, the peaks appeared at 2θ = 5.1° (*d* = 17.30 Å). In the BNCs film, these peaks are almost at the same position, indicating interlayer spacing increased in compared to Cloisite Na<sup>+</sup>/Tyr organoclay. This is evidence that these Cloisite Na<sup>+</sup>/Tyr are intercalated in PVA matrix, because for intercalated nanocomposites, the finite layer expansion associated with the polymer intercalation results in the appearance of a new basal reflection corresponding to the larger gallery height. On the other hand, for exfoliated structure, no more diffraction peaks are visible in the XRD diffractograms either because of a much too large spacing

**Fig. 3** TGA thermograms of **a** Cloisite Na<sup>+</sup>, Cloisite Na<sup>+</sup>/Tyr, **b** pure PVA and PVA/Cloisite Na<sup>+</sup>/Tyr BNC with different Cloisite Na<sup>+</sup>/Tyr contents **c** DTG of Cloisite Na<sup>+</sup>, Cloisite Na<sup>+</sup>/Tyr



**Fig. 4** FT-IR spectra of **a** Cloisite Na<sup>+</sup>/Tyr, **b** pure PVA and PVA/Cloisite Na<sup>+</sup>/Tyr BNCs with different Cloisite Na<sup>+</sup>/Tyr contents **c** 5%, **d** 10%, and **e** 15%

between the layers or because the nanocomposite dose not present ordering anymore [1].

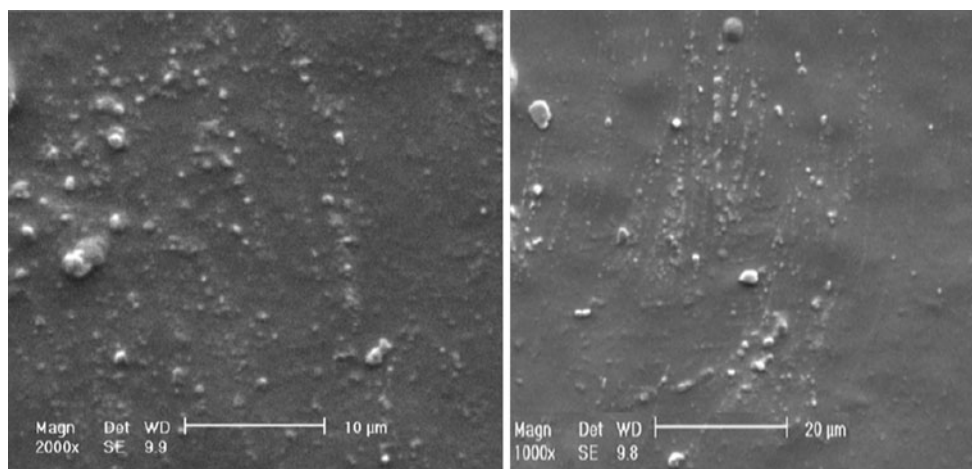
**Morphology**

The morphological micrograph of the BNC was studied by SEM, as shown in Fig. 5. The micrograph of PVA/Cloisite Na<sup>+</sup>/Tyr BNC film with 10% of organoclay is well and regularly dispersed in the PVA matrix. The white strands in the SEM images correspond to Cloisite Na<sup>+</sup>/Tyr platelets.

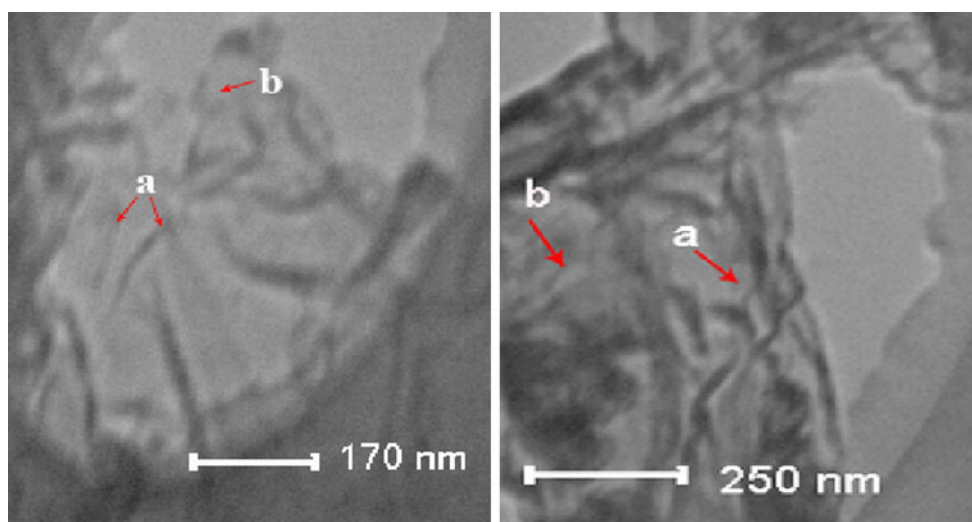
TEM images can give information about the morphology of BNCs. Typical TEM micrographs of the PVA/Cloisite Na<sup>+</sup>/Tyr BNC film with 10% of organoclay is illustrated in Fig. 6, which the dark lines are the layers of Cloisite Na<sup>+</sup>/Tyr. This TEM photograph demonstrates that most of Cloisite Na<sup>+</sup>/Tyr were intercalated and less exfoliated dispersed homogeneously into the PVA matrix. The clay layers were also shown to be well dispersed in the PVA hybrids by the XRD profiles, as discussed previously.

**Thermal stability of PVA/Cloisite Na<sup>+</sup>/Tyr BNCs film with different Cloisite Na<sup>+</sup>/Tyr contents**

All samples were dried in vacuum before TGA test. The TGA measurements of pure PVA and Cloisite Na<sup>+</sup>/Tyr BNCs with different Cloisite Na<sup>+</sup>/Tyr contents are shown in Fig. 3b. The thermal decomposition behavior of pure PVA and PVA/Cloisite Na<sup>+</sup>/Tyr BNCs with different percent of Cloisite Na<sup>+</sup>/Tyr passes through three steps when heated from 25 to 800 °C. In the first step, up to 200 °C, there is nearly no difference between the pure PVA and the PVA/Cloisite Na<sup>+</sup>/Tyr BNCs in thermal stability. The second decomposition temperatures of pure PVA, due to the elimination of side-groups at lower temperature, occurred at 200–325 °C and finally the third step around 600 °C due to the decomposition of main chain of PVA [35]. The pure PVA displays the lowest thermal stability



**Fig. 5** SEM micrograph of PVA/Cloisite Na<sup>+</sup>/Tyr (10%)



**Fig. 6** TEM micrograph of PVA/Cloisite Na<sup>+</sup>/Tyr (10%), **a** intercalated and **b** exfoliated

and major weight losses are observed in the range of, 200–500 °C for BNC films. The thermal stability of the BNCs film was increased with Cloisite Na<sup>+</sup>/Tyr content. The improvement of thermal stability can be explained through the reduced mobility of the PVA chains in the obtained BNCs. Because of reduced chain mobility, the chain transfer reaction will be suppressed and, consequently, the degradation process will be slowed and decomposition will take place at higher temperatures [36]. The temperature of 5 and 10% weight loss together with char yield at 700 °C for Cloisite Na<sup>+</sup>, modified Cloisite Na<sup>+</sup>, and PVA/Cloisite Na<sup>+</sup>/Tyr BNC with different Cloisite Na<sup>+</sup>/Tyr contents have been calculated from their thermograms. The thermal analysis data are summarized in Table 2. The char yield at 700 °C of the BNCs with different Cloisite Na<sup>+</sup>/Tyr content is higher than that of pure

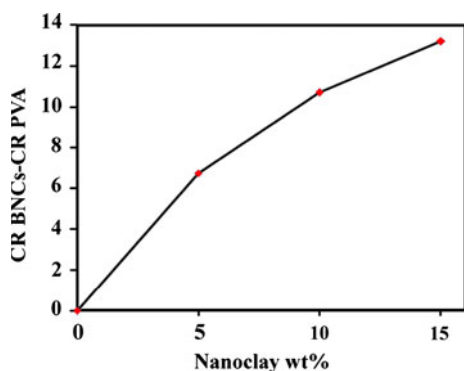
**Table 2** Weight loss (%) at different heating temperatures for the thermal decomposition of pure PVA and PVA/Cloisite Na<sup>+</sup>/Tyr BNCs with different Cloisite Na<sup>+</sup>/Tyr contents

PVA BNCs	Weight loss (%)		
	T <sub>5</sub> <sup>a</sup> (°C)	T <sub>10</sub> <sup>b</sup> (°C)	Char yield <sup>c</sup> (%)
Pure PVA	237	254	7.9
PVA/Cloisite Na <sup>+</sup> /Tyr (5%)	248	264	14.64
PVA/Cloisite Na <sup>+</sup> /Tyr (10%)	255	280	18.61
PVA/Cloisite Na <sup>+</sup> /Tyr (15%)	263	300	21.12

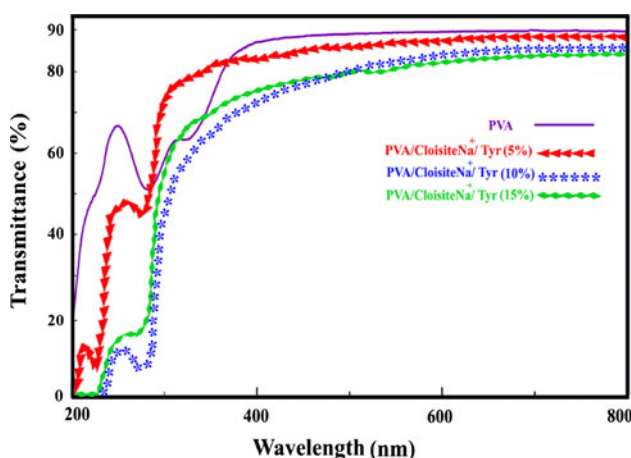
<sup>a</sup> Temperature at which 5% weight loss was recorded by TGA at heating rate of 10 °C min<sup>-1</sup> in N<sub>2</sub>

<sup>b</sup> Temperature at which 10% weight loss was recorded by TGA at heating rate of 10 °C min<sup>-1</sup> in N<sub>2</sub>

<sup>c</sup> Percentage weight of material left undecomposed after TGA analysis at maximum temperature 700 °C in N<sub>2</sub>



**Fig. 7** Char yield of nanocomposite—char yield PVA as a function of nanoclay wt%



**Fig. 8** UV-Visible transmittance spectra of PVA and PVA/Cloisite Na<sup>+</sup>/Tyr BNCs film containing 5, 10, and 15% Cloisite Na<sup>+</sup>/Tyr

PVA. This enhancement of the char formation is ascribed to the high heat resistance exerted by the Cloisite Na<sup>+</sup>/Tyr. The influence of Cloisite Na<sup>+</sup>/Tyr loading on the char formation of BNC films is shown in Fig. 7. Upon increasing the amount of Cloisite Na<sup>+</sup>/Tyr char yields of BNCs are enhanced.

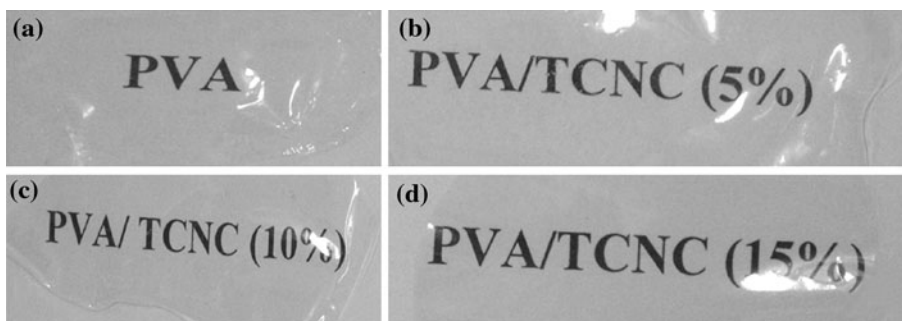
**Optical transparency**

Figure 8 shows UV–Visible transmission spectra of PVA and BNC films by different Cloisite Na<sup>+</sup>/Tyr content. In the visible light region (400–800 nm) the PVA shows about 90% transparency, but transparency of BNC films with increasing the amount of Cloisite Na<sup>+</sup>/Tyr were decreased. Owing to light scattering of clay nanolayer, UV regions (200–400 nm), are affected by the presence of the bionanoclay loading in the BNCs films. This can be explained by that Cloisite Na<sup>+</sup>/Tyr dispersed uniformly in nanoscale in the PVA matrix. Figure 9 also exhibits the optical transparency of PVA and BNCs film with different bionanoclay content. For this purpose the samples of BNCs film have been placed above a pattern and the photographs have been taken. The levels of transparency of these films are similar to that of pure PVA film and slightly affected by increasing the amount of organoclay loading.

**Conclusions**

Amino acids possess the potential to be used as modifiers, which can increase the interlayer spacing of clay and improve its biodegradability. Therefore, in this investigating chiral L-Tyr was used for the modification of Cloisite Na<sup>+</sup>. The intercalation of the Cloisite Na<sup>+</sup> was confirmed by XRD pattern. Then PVA/Cloisite Na<sup>+</sup>/Tyr BNC films were prepared by effectively dispersing of the inorganic nanolayers in PVA matrix by ultrasonic irradiation. BNCs of PVA are characterized by FT–IR spectroscopy, XRD, TGA, and TEM. Morphological images of PVA/Cloisite Na<sup>+</sup>/Tyr is studied through SEM. Thermal stability and optical clarity of PVA as well as PVA/Cloisite Na<sup>+</sup>/Tyr is also investigated by TGA and UV–Visible transmission spectra, respectively. The incorporation of nanolayers of modified Cloisite Na<sup>+</sup> in PVA matrix resulted in an increase in thermal decomposition temperature based on TGA. Furthermore, from UV–Visible data, although transparency of the obtained BNCs film was decreased

**Fig. 9** Transparency of PVA and BNCs film



upon increasing the amount of nanoclay, but it could be concluded that they still show comparable transparency to the pure PVA film.

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